CRYSTALLINE and QUASI-CRYSTALLINE INTERFACES

FROM ORDER TO DISORDER

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Characterization of interfaces

Homo-phase interface or **Grain Boundary** (GB)

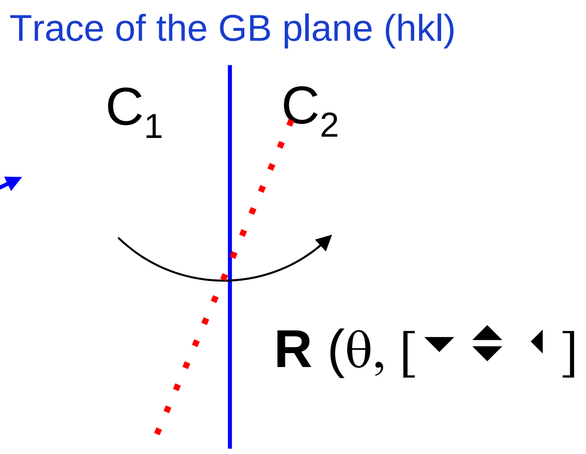
- Interface between two crystals of same nature and structure

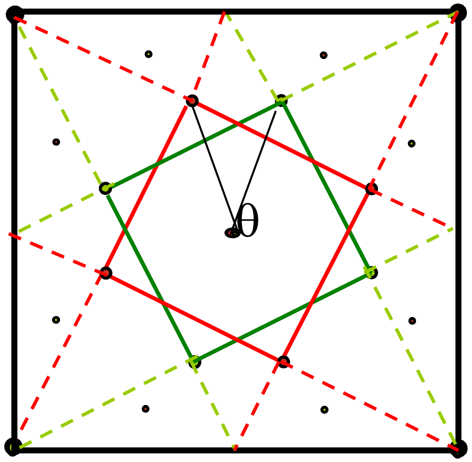
- characterized by a rotation **R (**θ [uvw]) or by a **coincidence index**

$\Sigma =$ □□□ρ

ρ : density of common nodes in the GB region

- And **a grain boundary plane** (hkl))

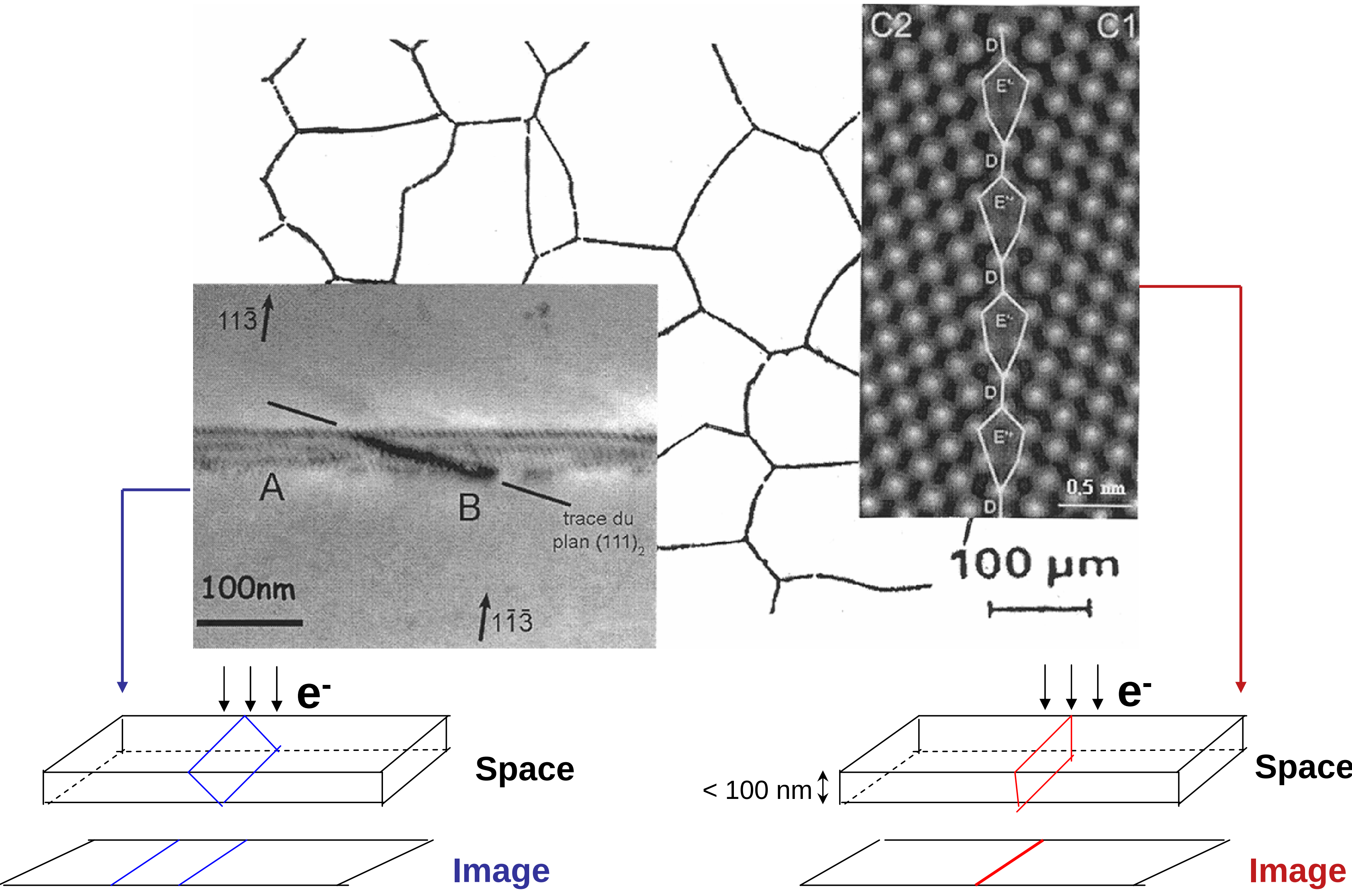




Hetero-phase interface **(interface)**

- Interface between two crystals of
  - different structure (two phases: for example f.c.c. / b.c.c. in iron)
  - different nature: metal/ceramic

A GRAIN BOUNDARY at DIFFERENT SCALES



EVOLUTION OF THE CONCEPT OF GB ORDER

1 - Amorphous cement *(W. Rosenhain and D.J. Ewen, J. inst. Metals 8* ***(1912)*** *149)*

2 - Periodic distribution of good fit and bad fit regions

W.T. Read and W. Shockley, Phys. Rev. 78 ***(1950)*** *275*

W. Bollmann, “Crystal defects and crystalline interfaces“, Springler, Berlin ***(1970)***

3 - Periodicity of structural units (SUs)

*A.P. Sutton and V. Vitek, Phil. Trans. R. Soc. Lond., A309 (****1983)*** *1 - 55*

4 - Quasi periodicity of structural units

*D. Gratias and A. Thallal, Phil. Mag. Letters, 57 (****1988)*** *63*

5 - Amorphous state of some GBs ?

*D. Wolf, Current opinion in Solid State and Materials Science 5 (****2001)*** *435.*

Outline

EVOLUTION OF THE CONCEPT OF GB ORDER

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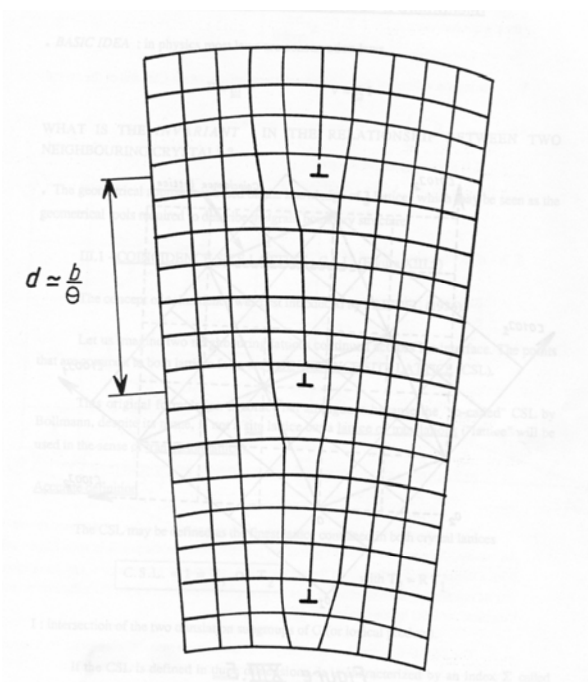
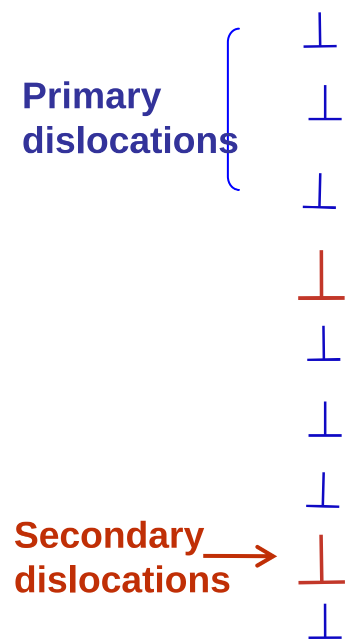
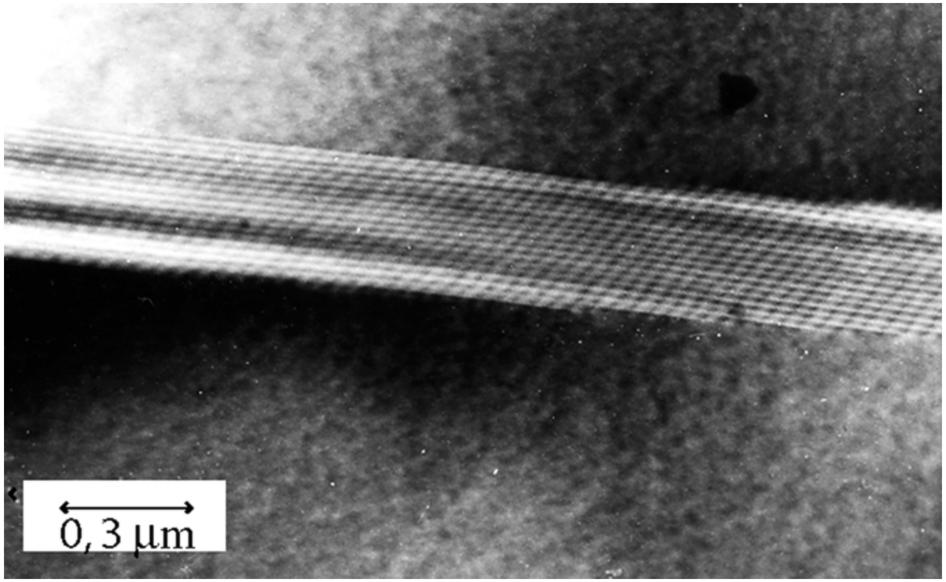
For low angle tilt GB



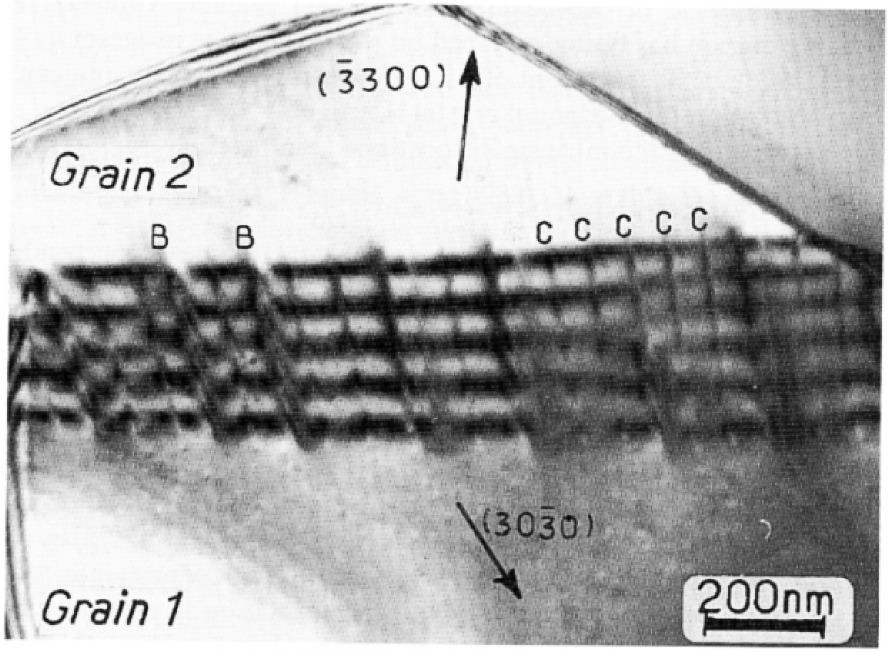
For any GB : intrinsic dislocations



Some examples of intrinsic dislocations



Primary intrinsic dislocations in low - angle (2°) grain boundary in a Fe-Mo alloy



Secondary intrinsic dislocations in a high-angle (85.5°) grain boundary in alumina (oxide)

OUTLINE

The structural unit model

Periodicity of structural units (SUs)

*A.P. Sutton and V. Vitek, Phil. Trans. R. Soc. Lond., A309 (**1983)** 1 - 55*

Quasi-crystalline interfaces

Quasi periodicity of structural units

*D. Gratias and A. Thallal, Phil. Mag. Letters, 57 (**1988)** 63*

Amorphous state of some GBs ?

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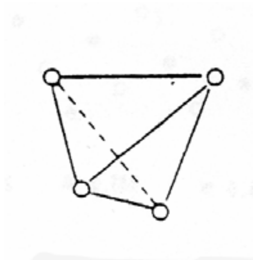
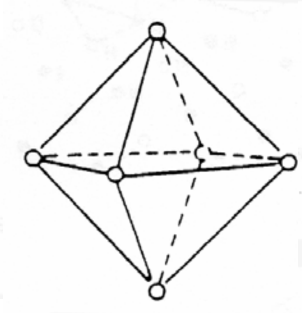
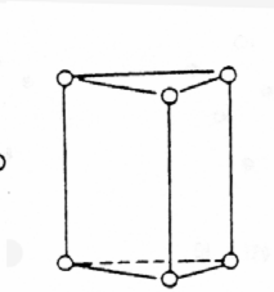
STRUCTURAL UNIT ≡ POLYHEDRAL CLUSTER OF ATOMS



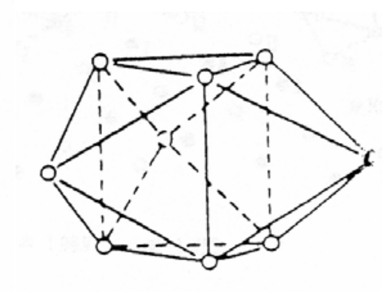
Tetrahedron



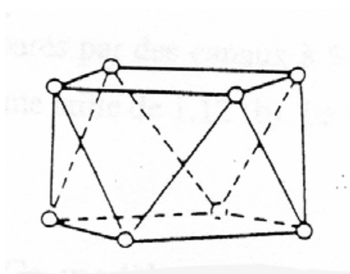
Octahedron



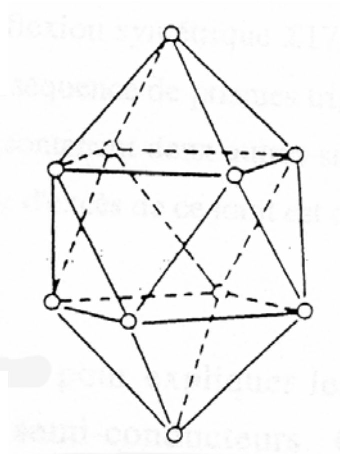
Trigonal prism



Capped trigonal prism



Archimedean square antiprism



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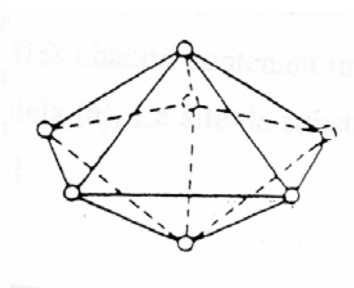


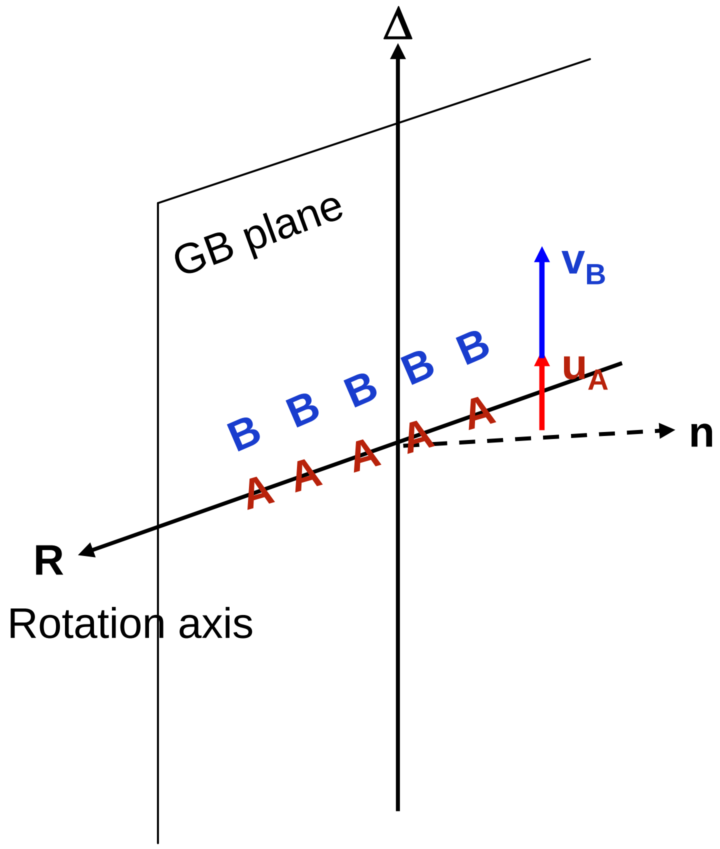
Pentagonal bipyramid

Equivalent to the elemental cells in crystals (cube, hexagon …)

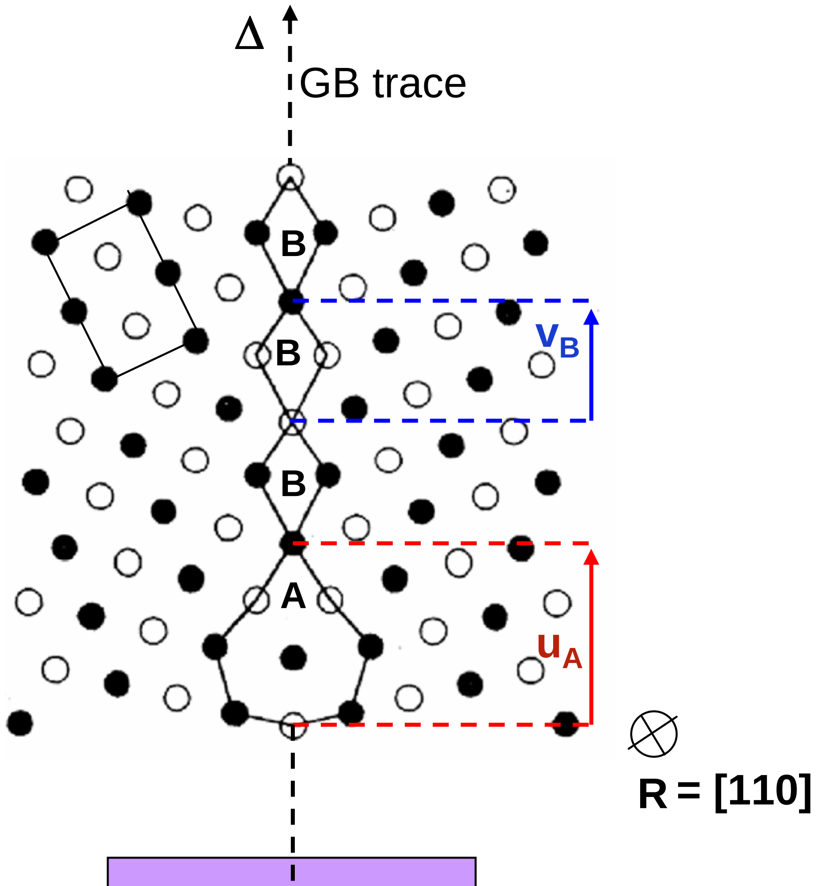
Limited number of polyhedra

Analogy with the hard sphere model of liquid structure - 5 similar clusters *(Bernal - 1964)*

STRUCTURAL UNIT MODEL GEOMETRY



$p = mu_A + n\, v_B$



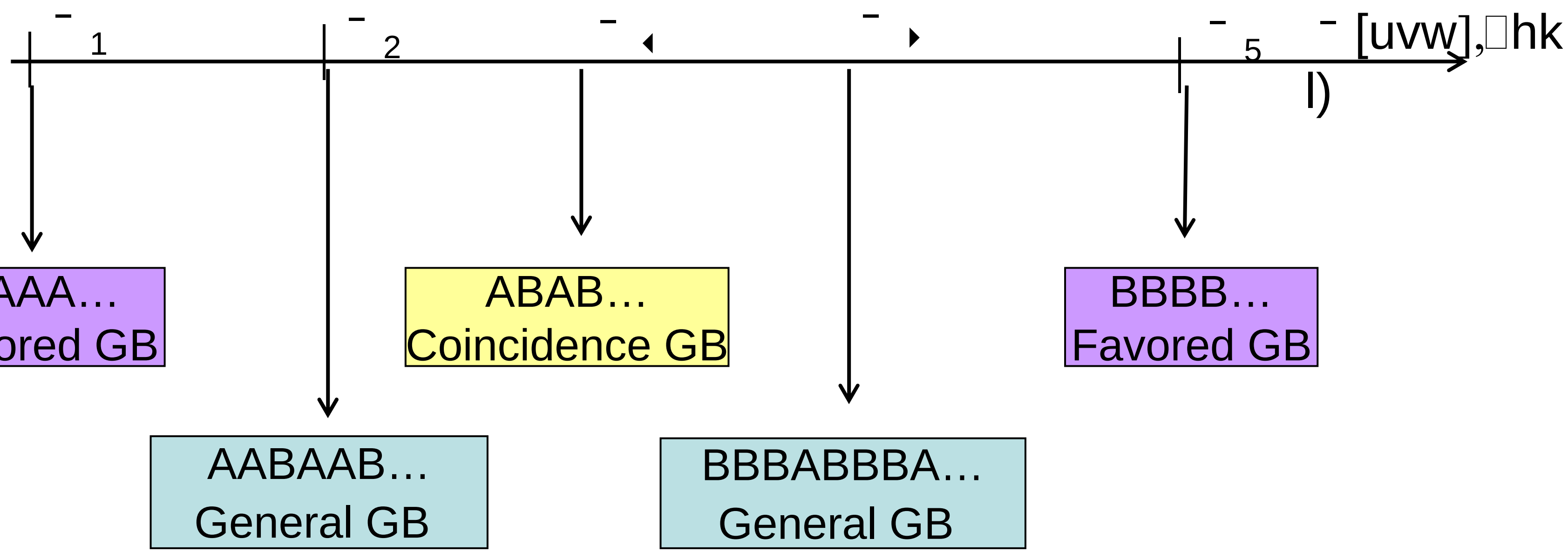
$p = 1u_A + 3\, v_B$

- Rational ratio m/n $\Rightarrow$ Periodic grain boundary

- Irrational ratio m/n $\Rightarrow$ Quasi-periodic grain boundary

STRUCTURAL UNIT MODEL PRINCIPLE

Any long period GB may be described as a sequence of structural units of two short period (favored) GBs.

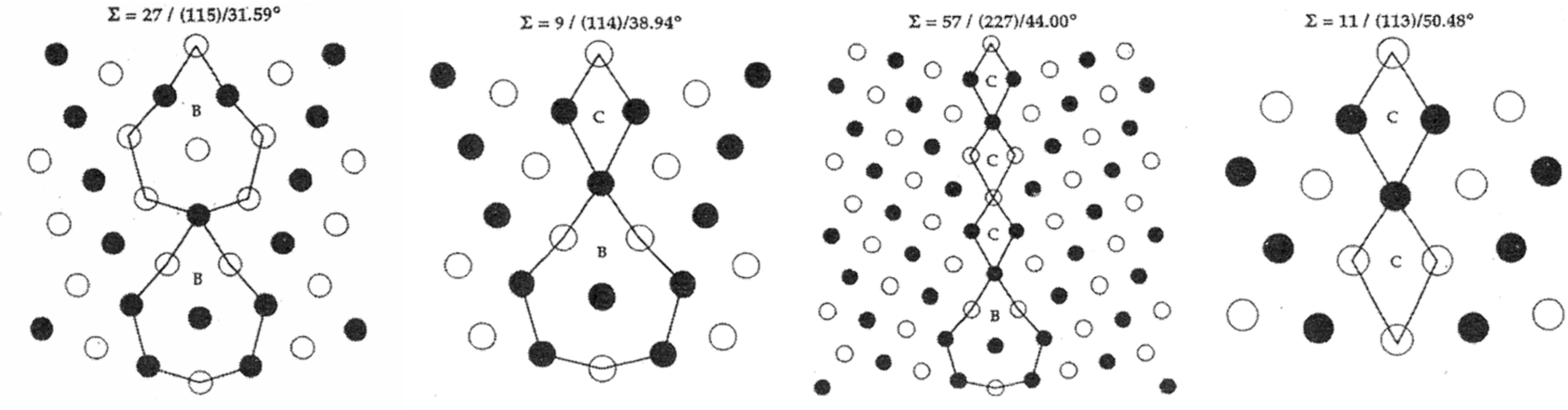


simple principle BUT COMPLEXITY → Multiplicity of descriptions

→ Distortion of the SUs

→ Hierarchy of descriptions

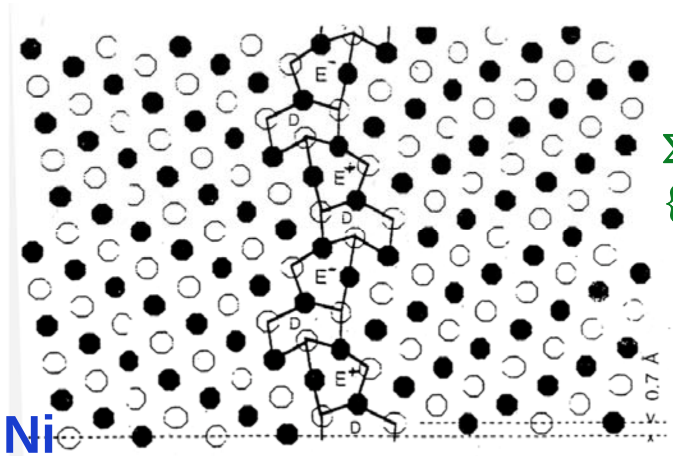
Series for symmetrical tilt GB around <110> for aluminium (FCC)



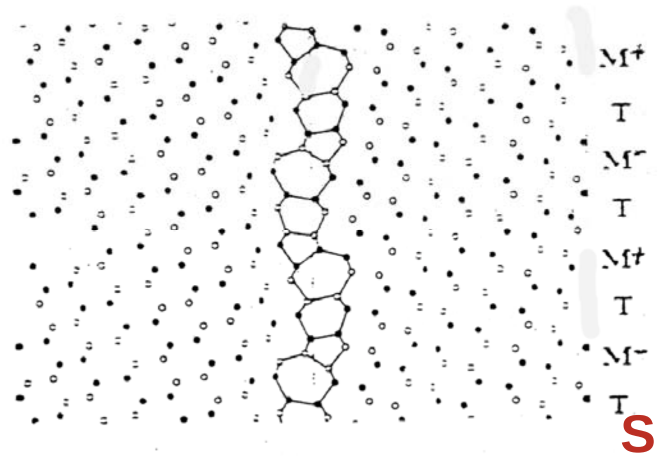
A given GB (same R and θ) in different materials

Period:

E$^+$ D E$^-$ D



$\Sigma = 11$
$\{332\}$

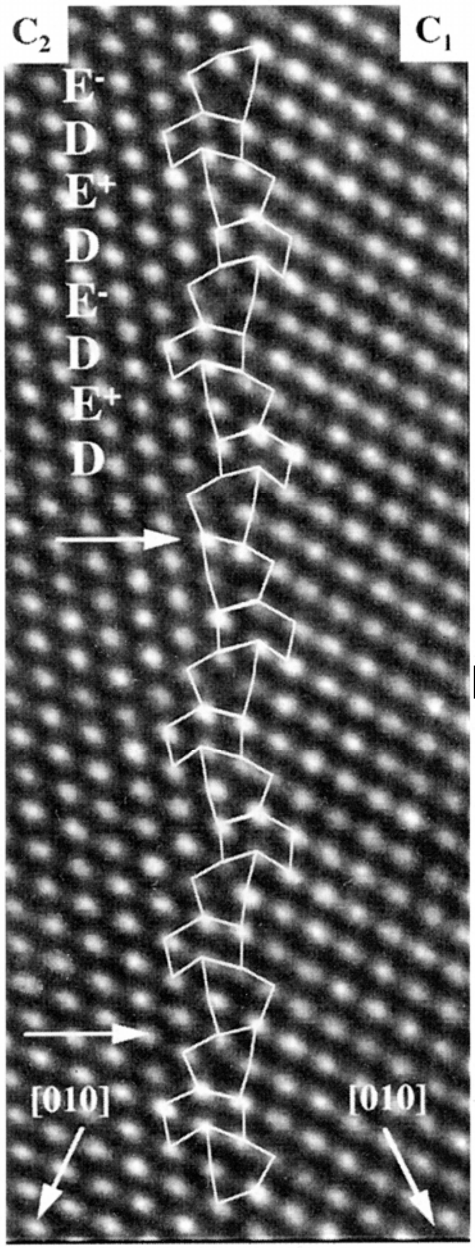


Period:

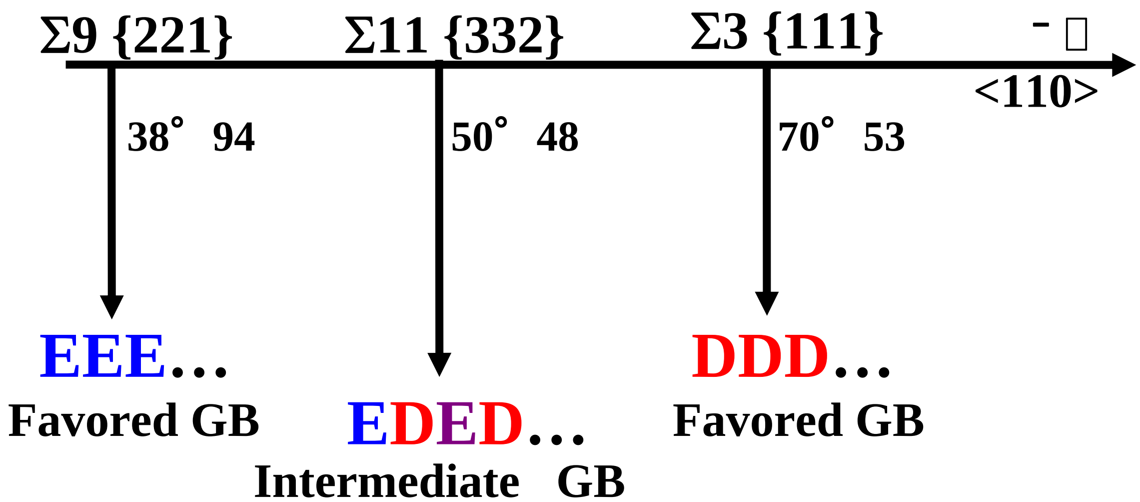
M$^+$ T M$^-$ T

The shapes of the structural unit differs but the period is similar

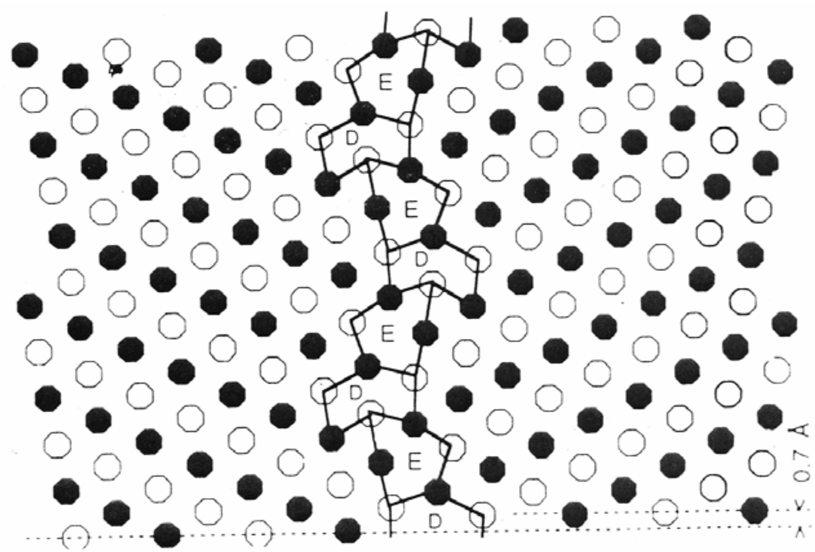
Description in terms of Structural Units (SU)



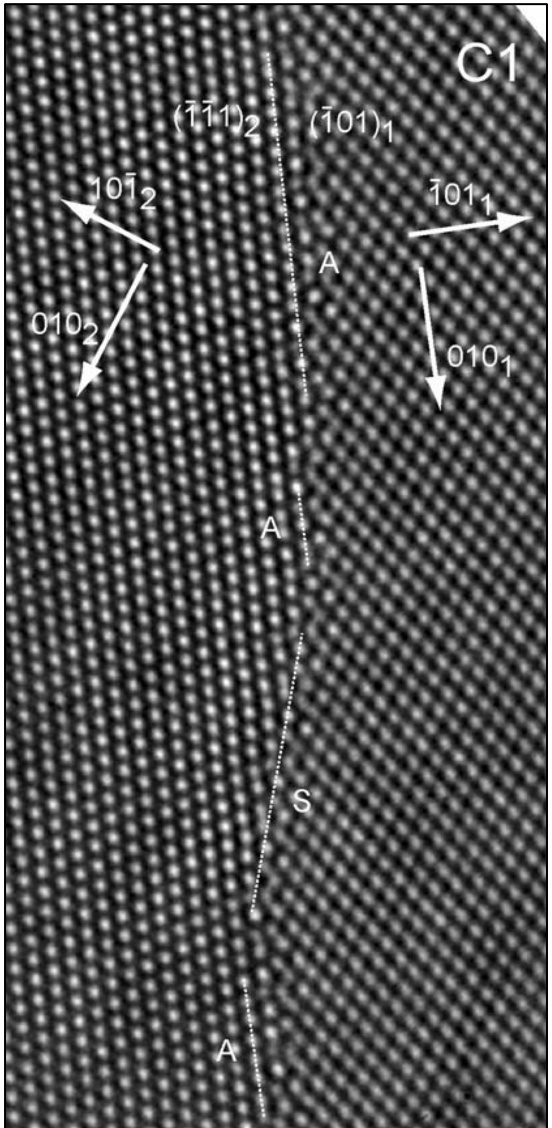
HRTEM image



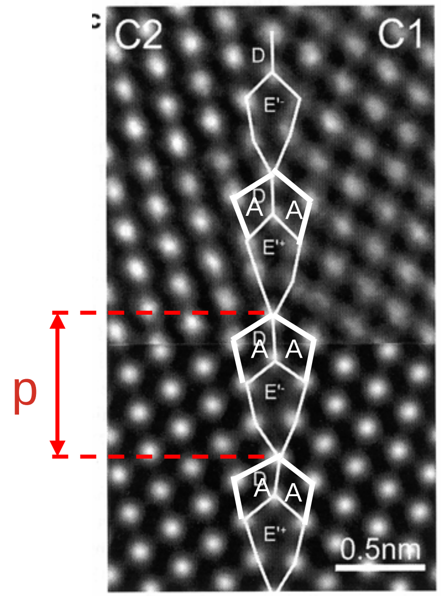
Simulation

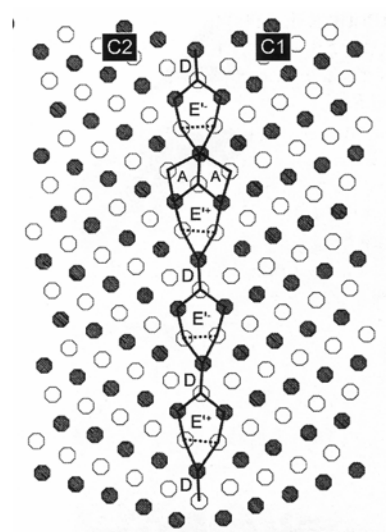


VALIDITY of SU MODEL for FACETTED GB - Near Σ9 (Cu)

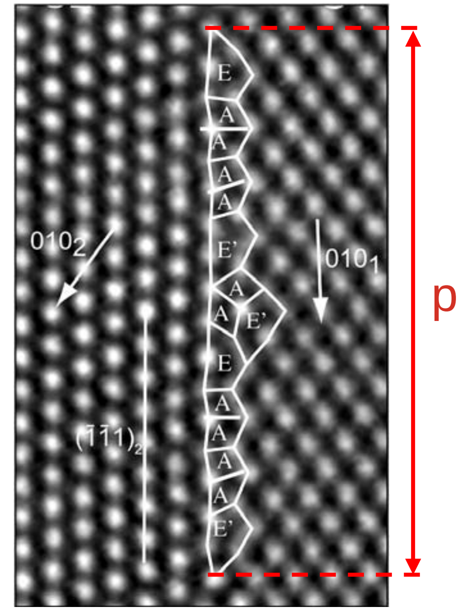


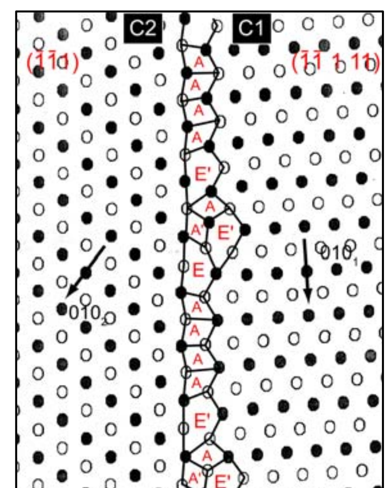
Symmetrical {221} facet





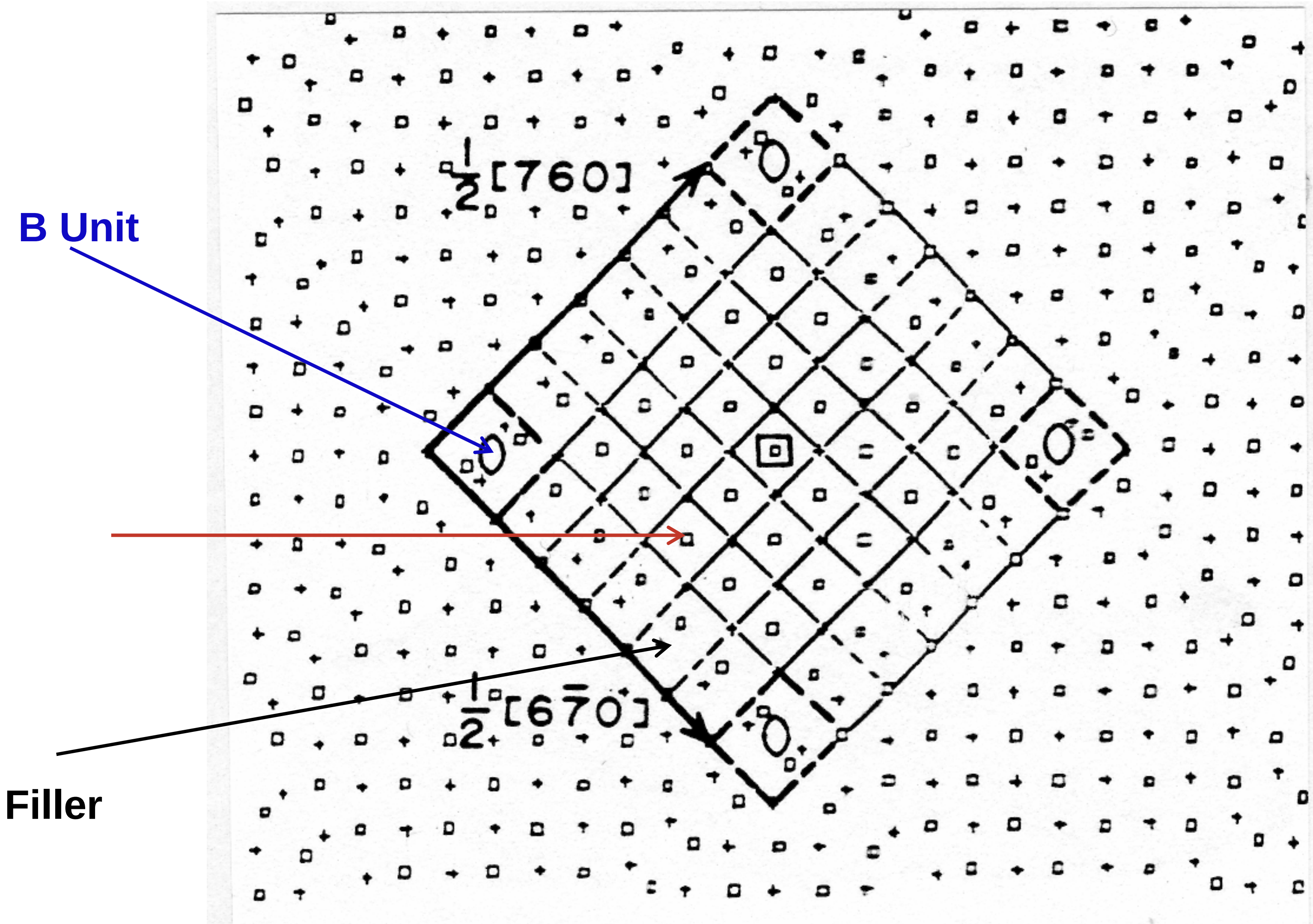
Asymmetrical facet



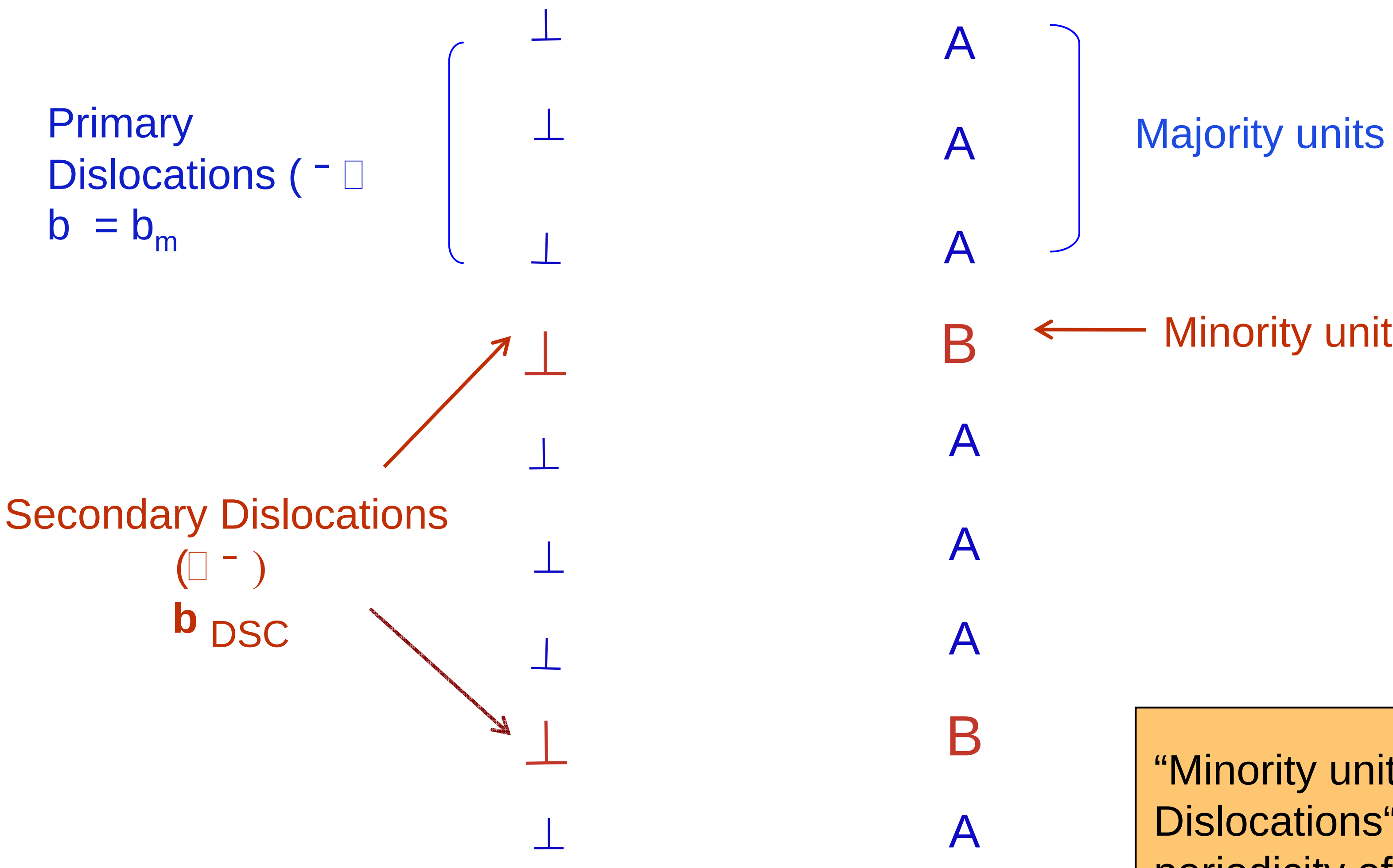


STRUCTURAL UNIT MODEL FOR TWIST GBs

Example of Σ 85 - 8.80°   [001]

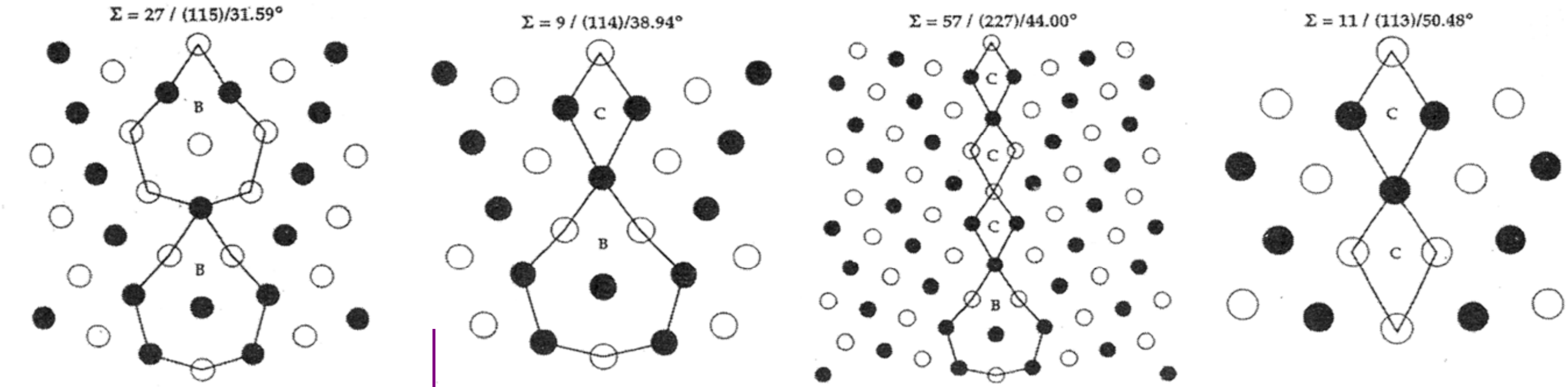


“STRUCTURAL UNITS/ INTRINSIC DISLOCATIONS

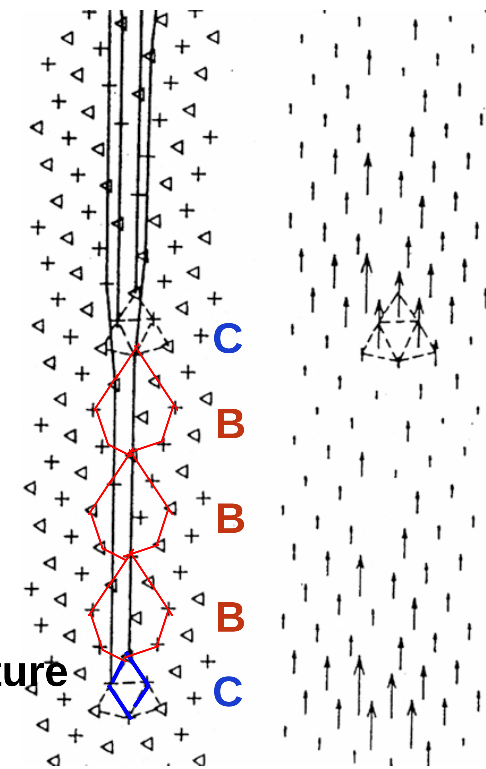


“Minority units/secondary Dislocations“ disturb the periodicity of “majority units /primary dislocations“

HRTEM IMAGES and HYDROSTATIC STRESS FIELDS



Σ = 89 (229)



Simulated structure

|BBBC|

Hydrostatic stress fields

Predicted structure

θ	Σ	Plan du joint	Structure
31.59°	27	(115)	|B.B|
34.89°	89	(229)	|□□□C|
38.94	9	(114)	|□C|
50.48°	27	(113)	|C.C|

STRUCTURAL UNIT MODEL : Multiplicity of descriptions

A favored GB may be described by differents SUs whose the energies are very

⇓

Any intermediate GBs may be constituted by different combinations N of these

Favored GB
i possible SUs: A, A', A"

Favored GB
j possible SUs: B, B'

GB period : $\mathbf{p} = m\,\mathbf{u}_A + n\,\mathbf{v}_B$

$$N = i^m j^n$$

All the N configurations are not stable

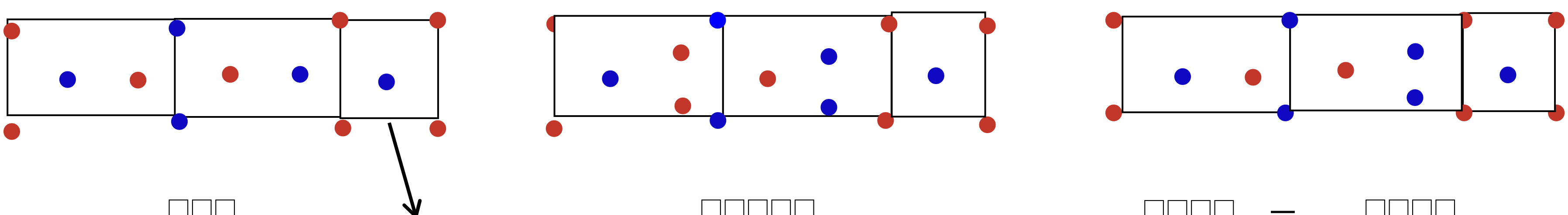
Comparisons with the hydrostatic stress field and with the HRTEM images

Examples of multiplicity of descriptions

Favored tilt GB □ 5 (210) - 36.9° [001]

□ □□

Coincidence GB □ 17 (530) – 28.1 ° [001]



□□□ □□□□□ □□□□ ≡ □□□□

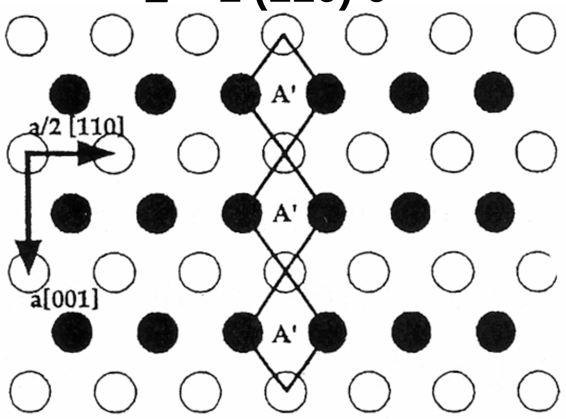
A: perfect crystal

$N = 2^2.1^1 = 4$

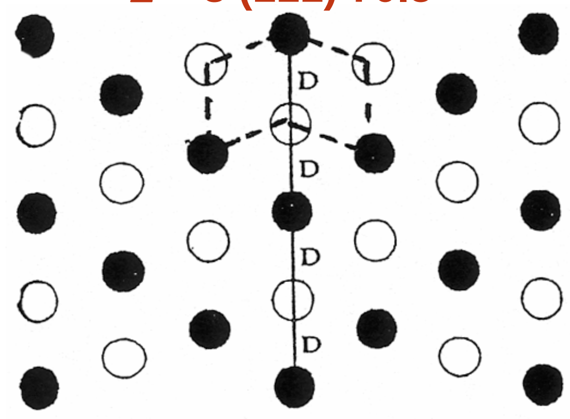
Energy ratio: 1.07 / 1.09/ 1

STRUCTURAL UNIT DISTORTION

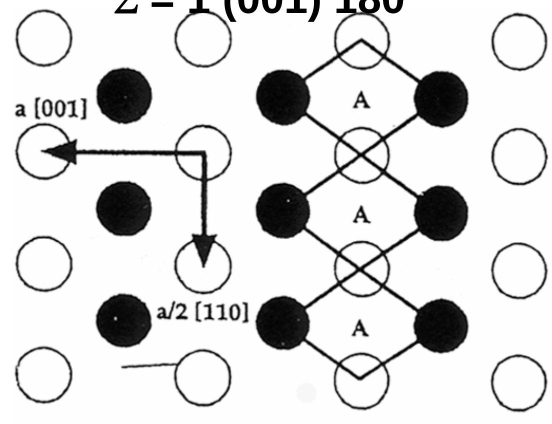
$\Sigma = 1$ (110) 0°



$\Sigma = 3$ (111) 70.5°



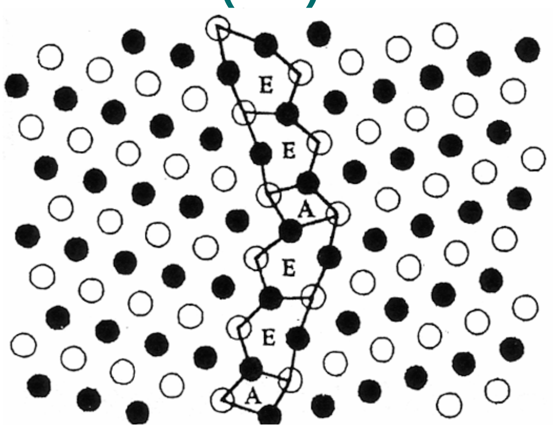
$\Sigma = 1$ (001) 180°



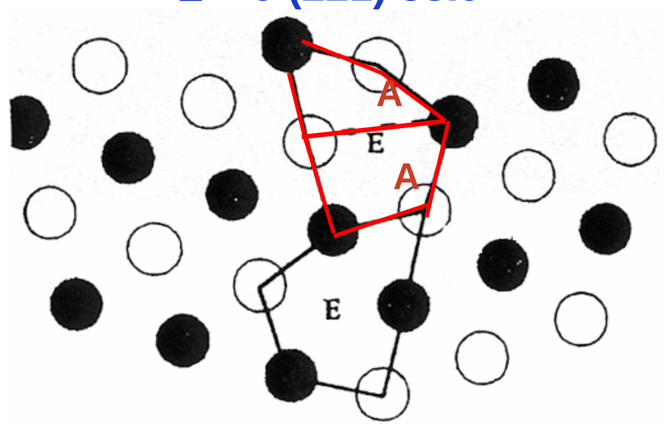
Σ 1 (single crystal): same unit A and A' rotated by 18

Σ = 3 (twin): unit D ≡ 2 A units rotated by 70.5°

$\Sigma = 27$ (552) 32.5°



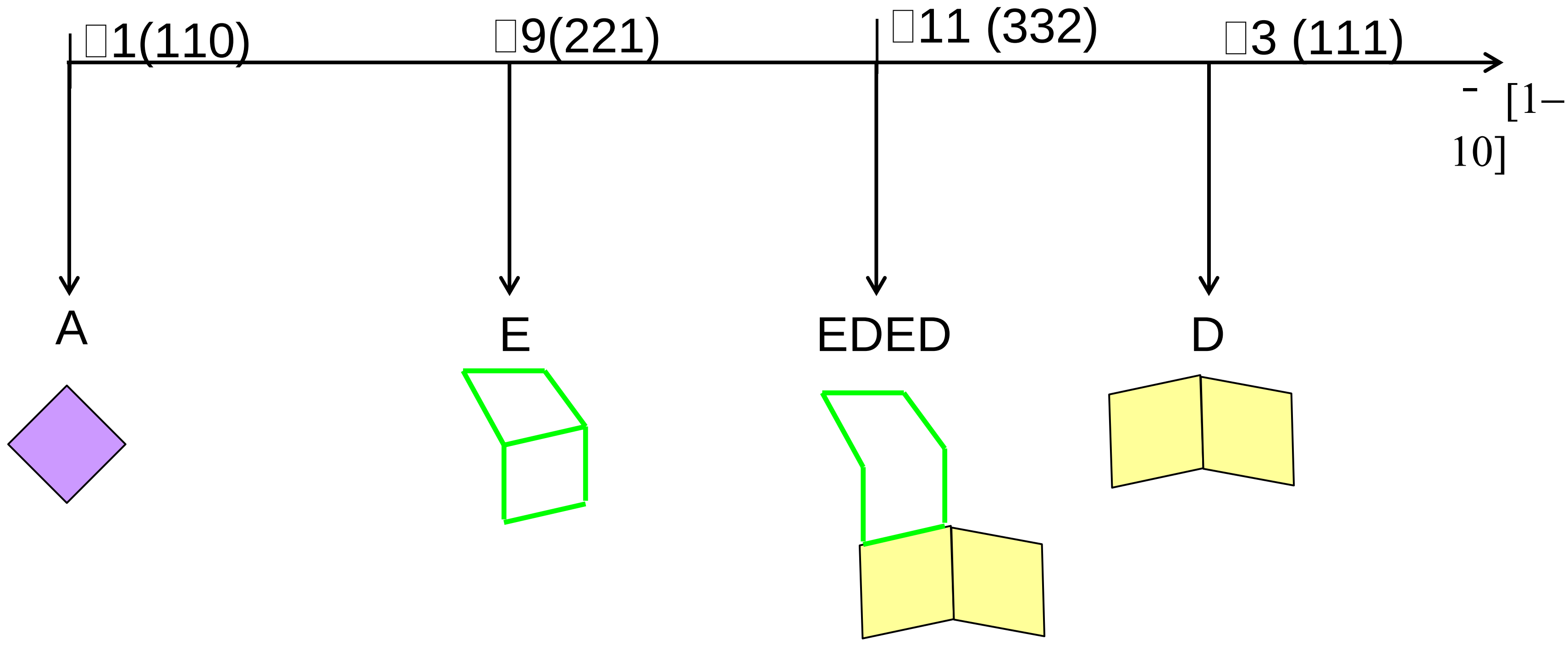
$\Sigma = 9$ (221) 38.9°



Σ = 9: unit E formed by two distorted and rotated A units

Σ = 27: period = EEA but some E units are distorted

SU DISTORTION ⇒ HIERARCHY of GB DESCRIPTIONS



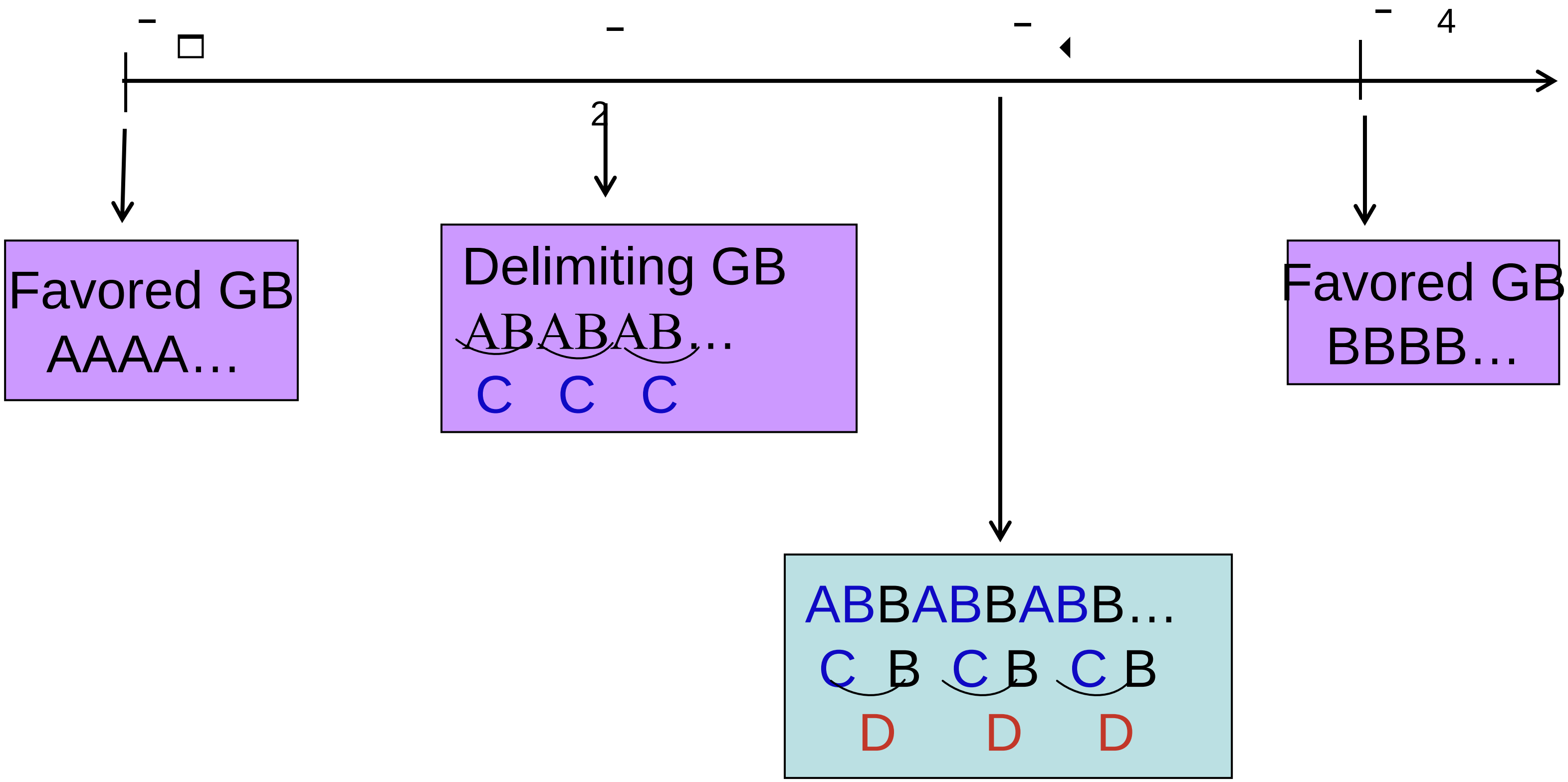
□9(221) could be described by A and D units but strong **distortion**

⇓

Better description by E unit ⇒ then use of E for the structure of □11 (332)

□9 appears as a delimiting GB

HIERARCHY OF GB DESCRIPTIONS



General rational GBs
(rational ratio m/n of A and B units)

- As the order of the description increases $\Rightarrow$ the distortions of the SUs decrea
- The atomic description requires the knowledge of the basic structures

HOW TO GENERATE the SEQUENCE of SUs ?

$$\underbrace{\mathbf{p} = m\mathbf{u}_A + n\,\mathbf{v}_B}$$

There is a huge number of ways for arranging m units A and n units B in a periodic fashion

$$W = \frac{(m + n - 1)!}{m!\, n!}$$

(For m = 13 and n = 19, W = 10.855.425)

THUS

To determine the sequence of structural units, it is necessary to use:

- **an algorithm**

A.P. Sutton and V. Vitek, Phil. Trans. R. Soc. Lond., A 309 (1983) 1.

Main assumption: The boundary structure changes in as smooth and continuous manner as possible when θ varies

- **a strip band method** (analogous to what is used for quasicrystallography),

A.P. Sutton, Prog. Mat. Sci. 36 (1992) 167.

ALGORITHM to DETERMINE THE S.U. SEQUENCE in a GB

Principle - GB period = mA + nB with (m < n)

$$\frac{1}{p+1} < \frac{m}{n} < \frac{1}{p}$$

$\underbrace{A\,(p+1)\,B}_{C} \Leftarrow X \Rightarrow \underbrace{A\,pB}_{D}$

$X \rightarrow r\,C + s\,D$

$r = n - mp$
$s = m\,(p+1) - n$

$$\frac{1}{q+1} < \frac{r}{s} < \frac{1}{q}$$

$\underbrace{C\,(q+1)\,D}_{E} \Leftarrow X \Rightarrow \underbrace{C\,qD}_{F}$

$X \rightarrow t\,E + v\,F$

and so until the number of minority units = 1

Ex: Σ = 4233 (16,16,61) - period = 13 A + 19 B

$$\frac{1}{2} < \frac{13}{19} < \frac{1}{1}$$

$\underbrace{ABB}_{C} \Leftarrow X \Rightarrow \underbrace{AB}_{D}$

$r\,C + s\,D$

$r = 6\ ;\ s = 7$

$$\frac{1}{2} < \frac{6}{7} < \frac{1}{1}$$

$\underbrace{CDD}_{E} \Leftarrow X \Rightarrow \underbrace{CD}_{F}$

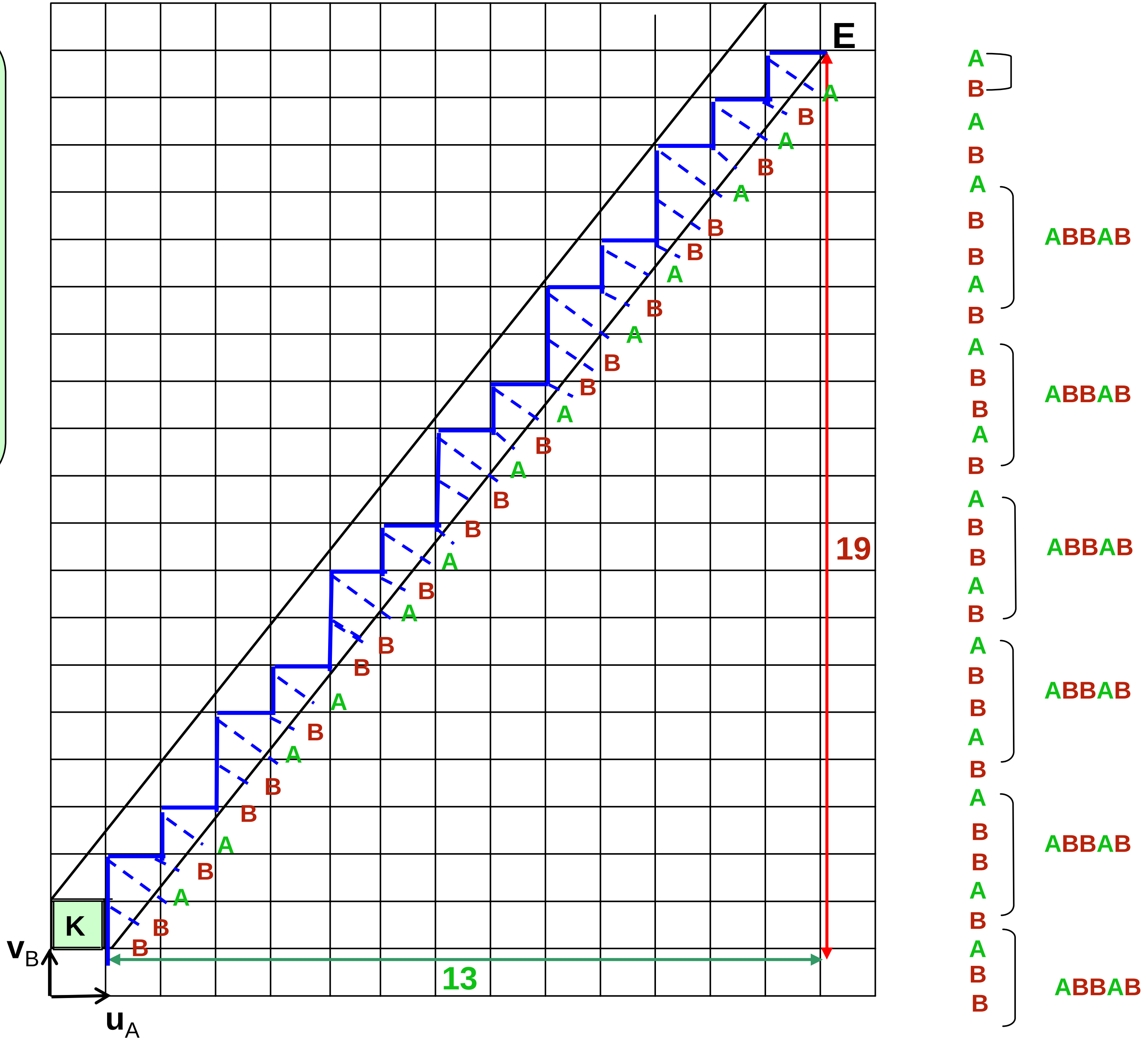
$1\,E + 5\,F$

$1\,E + 5\,F \equiv CDD - 5\,CD \equiv D - 6\,CD \equiv AB - 6\,(ABBAB)$

SU sequence: ABABBABABBABABBABABBABABBABABBAB

The algorithm always results in the largest distance as possible between the minority units
Two adjoining minority units never appear

STRIP METHOD for determining the SU sequence in a GB



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Quasi-crystalline interfaces

Quasi periodicity of structural units

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Amorphous state of some GBs ?

*D. Wolf, Current opinion in Solid State and Materials Science 5 (**2001**) 435.*

HOW TO GENERATE QUASIPERIODIC SEQUENCES

ALGORITHM *(Levine and Steinhard, 1984)*

For irrational tilt GBs: $m_A / n_B = \underbrace{m / n}_{\text{rational}} + \underbrace{\lambda}_{\text{irrational}}$

More simple quadratic form such as:
$\lambda^2 - \lambda - 1 = 0$ *in that case*
$\lambda_1 = \underbrace{\tau = (1+\sqrt{5})/2}_{\text{Golden number}}$

$u_A = \begin{pmatrix} 1 \\ 0 \end{pmatrix}$ $v_B = \begin{pmatrix} 0 \\ 1 \end{pmatrix}$ Self - similar sequence obtained by applying the operation $M = \begin{pmatrix} 1 & 1 \\ 1 & 0 \end{pmatrix}$

Then repeat…

(det $M = -1$)

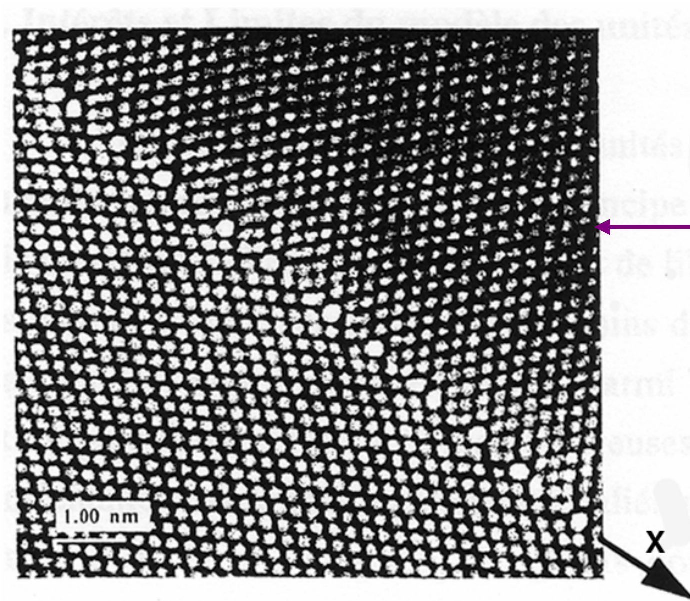
Number of iterations	Sequence of US	m_A / n_B
0	AB	1/1
1	BAB	1/2
2	BABBA	2/3
3	BABBABAB	3/5
4	BABBABABBABBA	5/8
⇓	⇓	⇓
∞	**Quasi-periodicity**	**1/ τ**

Quasiperiodic GBs are the limits of periodic GBs with increasing periods

STRIP METHOD

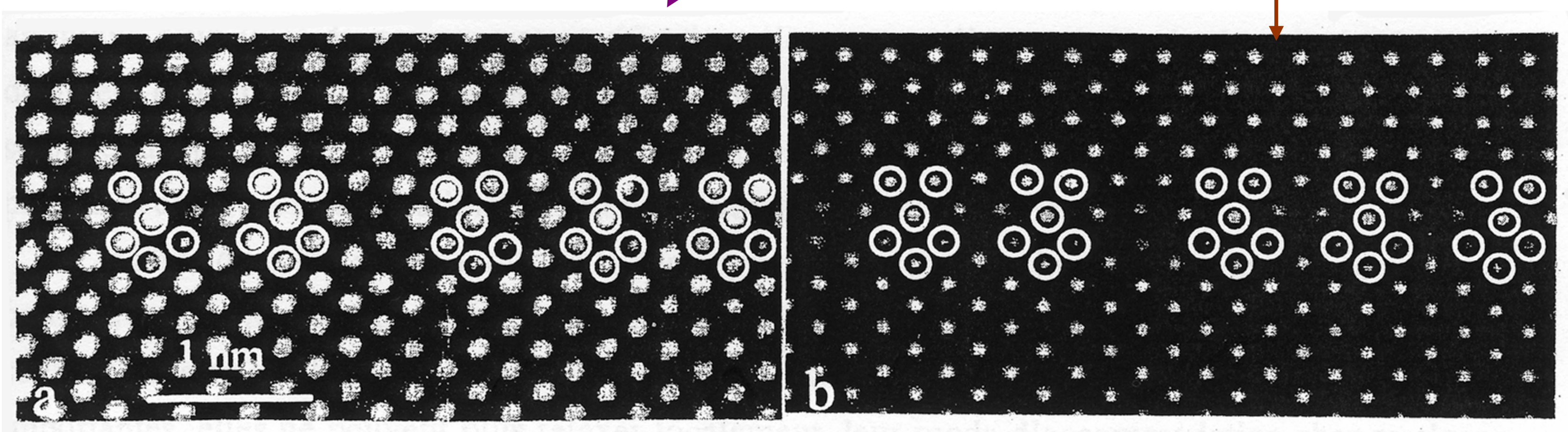
Irrational slope of the E line in the section/projection method

Quasiperiodic Structure of a GB in gold



□R□□□□image of a quasi periodic GB
90° [001] $(100)_I$ // $(110)_{II}$

Simulated image based on calculation of the GB structure
(*approximant 29/ 41* ε *= 3. 10* $^{-4}$)



J.M. Pénisson and al., Mat.Sci. Forum, Trans Tech Pub. 294-296 (1999)

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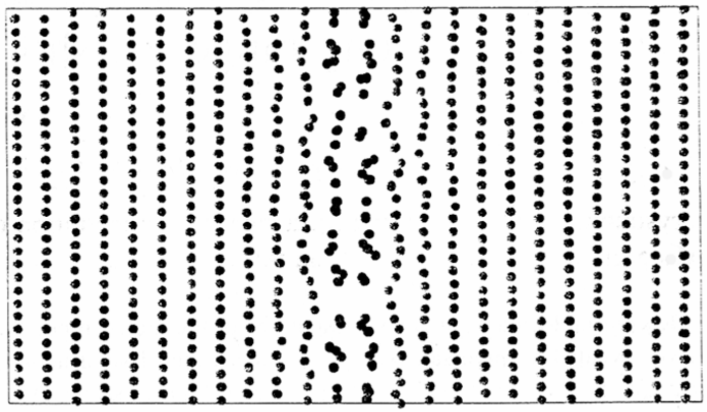
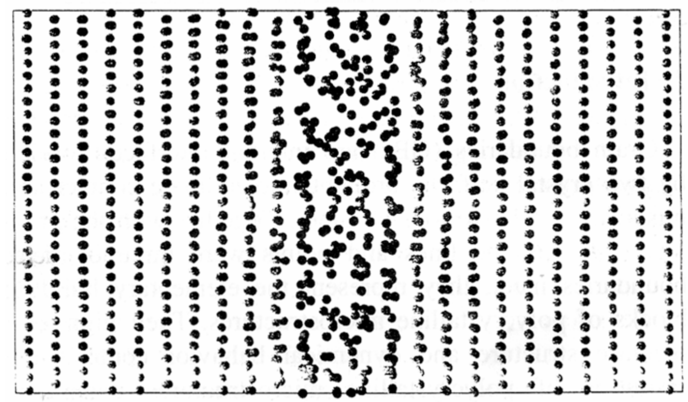
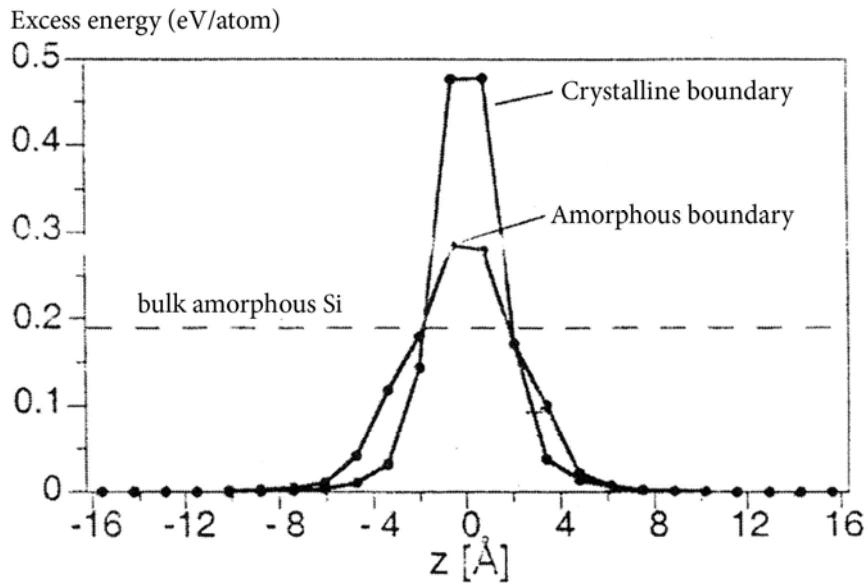
CRYSTALLINE / AMORPHOUS STATE of a $\Sigma = 29$ TWIST GB in SILICON



Crystalline GB *(relaxed at low T°)*



Amorphous GB *(relaxed at high T°)*



(c)

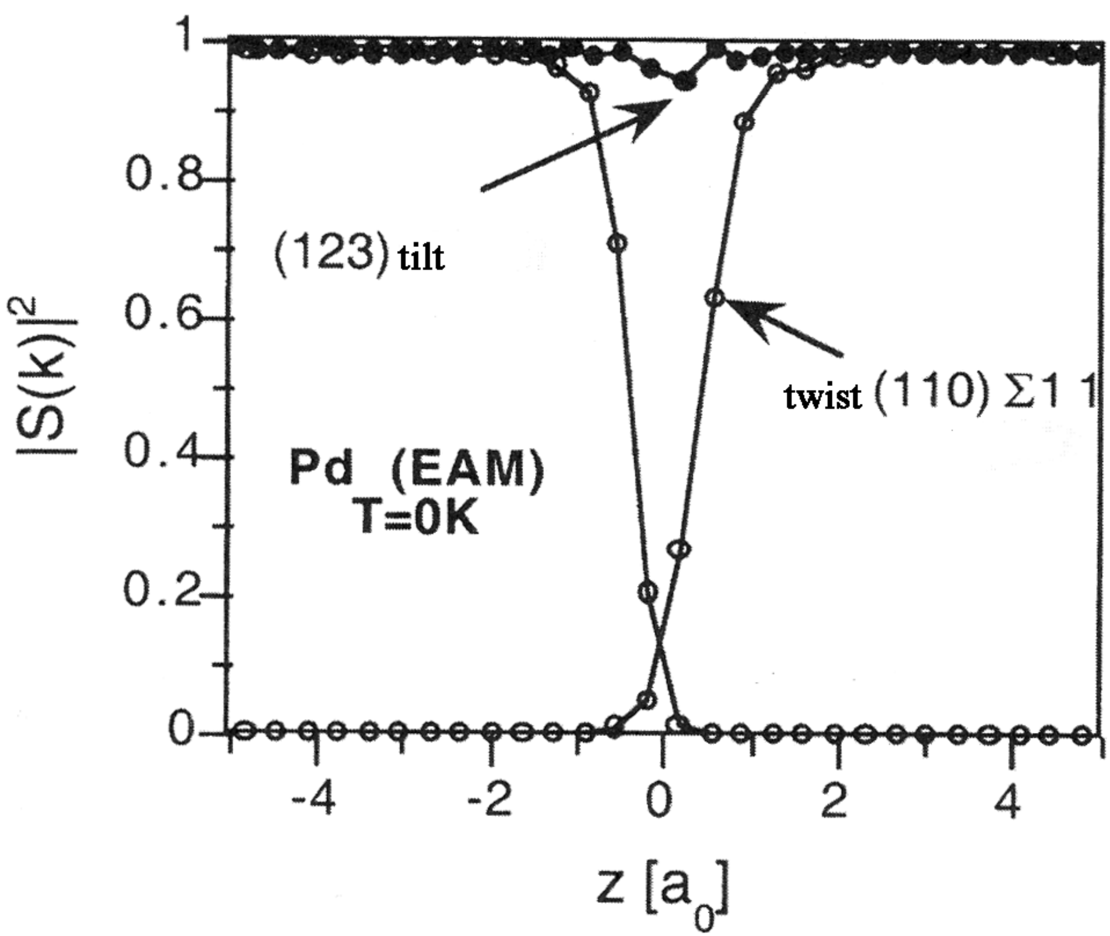
D. Wolf, Current opinion in Solid State and Materials Science 5 ***(2001)*** *435*

GB ORDER / DISORDER ?

Distinction between ORDER and ENERGY

ENERGY → not controlled by the order at large distances (periodicity))

ENERGY → controlled by the short-distance order or local arrangement of atoms (



The square of the structure factor S (k) ^{2} is function of the crystallinity = 1 (if 100% crystal)

Tilt GB is crystalline

Twist GB is amorphous

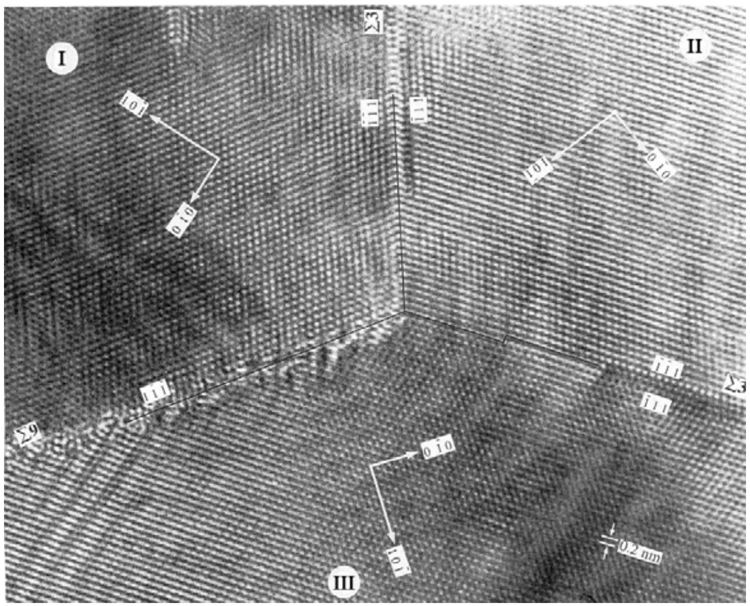
Although

$E_{(110)\ twist} < E_{(123)\ twist}$

REAL GRAIN BOUNDARIES

GBs are not infinite but connected to others in polycrystals

They are constrained at triple junctions

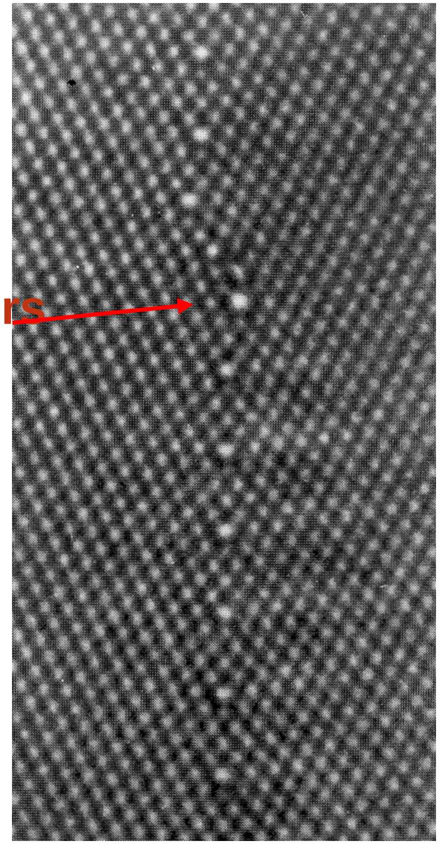


L. Priester, D.P. Yu,
J. Mat. Sci. Eng., A 188 (1994) 113.

GBs are not perfect

⇓

they contain defects



Knowledge of defects are fundamental for GB properties, then polycrystal behaviou